

Experimental Measurement of the Temperature Rise Generated During Dynamic Crack Growth in Metals

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Abstract

During the high speed propagation of cracks, large temperature increases occur at the crack tip due to the intense dissipation of plastic work there. This increased temperature may have a significant effect on the material's dynamic fracture toughness. An experimental investigation of the temperature fields at the tip of dynamically propagating cracks in 4340 steel was performed using a focused array of high speed, infrared detectors. Temperature fields were measured for cracks growing at speeds from 700 m/s to 1900 m/s. Maximum temperature increases were as high as 465°C. The temperature fields were differentiated to determine the plastic work rate distribution at the crack tip and to estimate the plastic strain rate. Effects of crack tip heating on dynamic fracture toughness are discussed.

Riassunto

Durante la propagazione ad alta velocità delle cricche, all'apice di queste ultime si verificano forti aumenti di temperatura dovuti ad un'intensa dissipazione locale di lavoro plastico. Tale aumento di temperatura può avere un significativo effetto sulla tenacità di frattura dinamica del materiale. Un'indagine sperimentale dei campi di temperature all'apice delle cricche a propagazione dinamica sull'acciaio 4340 è stata effettuata con l'impiego di una batteria focalizzata di rivelatori IR ad alta velocità. I campi di variazione di temperature sono stati misurati per cricche propagantisi a velocità di 700-1900 m/s. I massimi incrementi di temperatura arrivavano fino a 465°C. I campi di temperatura sono stati differenziati per determinare la distribuzione delle velocità di lavoro plastico in corrispondenza dell'apice delle cricche e per valutare la velocità di deformazione plastica. Vengono discussi gli effetti prodotti dal riscaldamento all'apice delle cricche sulla tenacità della frattura dinamica.

Introduction

It was recognized by Krafft and Irwin [1] that the heat generated at the tip of a dynamically propagating crack may affect the dynamic fracture propagation toughness of a material. The source of the crack tip heating during dynamic crack growth in metals is the dissipation of plastic work near the crack tip. Taylor and Quinney [2] and others, see Bever, et al. [3], have shown that most of the energy of plastic deformation (>90%) is dissipated as heat while the remainder is stored in the material in the form of dislocation energy, and energy due to other defects created by the deformation.

Despite the potential importance of crack tip heating on dynamic fracture toughness no attempt to directly measure the temperature at the tip of a crack in a metal has yet been made. Recent advances in infrared detectors now make it possible to measure, in real time, the spatial distribution of crack tip temperature for dynamically propagating cracks.

Such experiments are described here and results for the distribution of temperature and of plastic work rate near the crack tip are discussed. The concept of the experiment is as follows: A linear array of eight high speed, indium antimonide infrared detectors is used to measure the temperature at eight observation points covering a line segment perpendicular to and intersecting the path of a crack propagating dynamically in a specimen of 4340 steel. As the crack traverses the line segment, the voltage outputs from each of the eight detector elements are recorded simultaneously on a set of digital oscilloscopes. At the same time the crack length record is measured using a conductive grid technique. Knowing the temperature versus time at the observation points and using the measured crack length versus time record, the temperature distribution near the crack tip is found. This information is used to assess the role of crack tip heating on fracture toughness, and to determine the plastic work rate. This experimental setup is similar to that developed by Duffy and co-workers [4,5] for studying dynamic shear bands.

Description of experiments

Electromagnetic radiation is constantly emitted from bodies that are at temperatures greater than

absolute zero. Since the radiation intensity increase as θ^4 , where θ is the absolute temperature of the body, measurement of radiation can provide a sensitive measurement of temperature. This is the central concept of the experiment described here. As the crack propagates, it heats up the material near the crack tip, causing infrared radiation to be emitted from the specimen surface. By measuring this radiation the temperature may be determined.

Test Specimens

The specimens used for this experiment were wedge-loaded compact tension specimens of 4340 steel with the dimensions shown in Fig. 1. The mechanical and thermal properties and chemical composition are given in Table 1. The specimens were prepared by grinding, lapping, and then polishing the specimen surfaces with a 9 μ m diamond paste. Several specimens had side grooves machined along the crack paths. Two side groove geometries were used, a semicircular groove of radius 1.5 mm, and a V-notch groove of width 3 mm and depth 1.5 mm.

TABLE 1 - Mechanical and thermal properties and chemical composition 4340 steel, aircraft quality, vacuum degassed

Heat Treatment	σ_0 , MPa	K_{IC} , MPa \sqrt{m}	Hardness, HRC								
Quenched and tempered											
871°C 1 hr., air cool	1490*	50*	44-45								
843°C 1½ hr., oil quench											
316°C 1 hr., air cool											
Oil quenched											
871°C 1 hr., air cool	1700**	44**	49-55								
843°C 1½ hr., oil quench											
Chemical Composition, %											
C	Mn	P	S	Si	Cu	Ni	Cr	Mo	Sn	Al	Fe
.42	.71	.007	.013	.23	.12	1.78	.83	.25	.005	.033	Balance
Properties at 0°C:**											
Thermal conductivity	$k = 34.6 \text{ W/m} - ^\circ\text{K}$										
Thermal expansion	$a = 11.2 \times 10^{-6} \text{ } ^\circ\text{K}^{-1}$										
Specific heat	$c = 448.0 \text{ J/kg} - ^\circ\text{K}$										
density	$\rho = 7834 \text{ kg/m}^3$										
Thermal diffusivity	$\alpha = .99 \times 10^{-5} \text{ m}^2/\text{s}$										
* Source: Zehnder and Rosakis (1989)											
** Source: Aerospace Structural Metals Handbook, Metals and Ceramics Information Center, Battelle Columbus Laboratories, Columbus, Ohio, 1989.											

An array of lines of conductive paint spaced 3 mm apart was silkscreened onto the back of each specimen, using techniques developed by Hudak et al. [6]. As the crack runs, the conductive strips are broken, providing a time record of the history of the crack motion. The crack length record is differentiated using the incremental polynomial fit method described by Zehnder and Rosakis [7] to find the crack speed.

The specimens were loaded with a wedge driven slowly between the loading pins until a crack

was initiated from the initially blunted notch tip. Due to the geometry of the loading and the initial bluntness of the notch, the resulting crack propagates unstably at high speeds ranging from 2000 m/s to 1000 m/s, depending on the initial bluntness.

Infrared detectors

Based on analytical results of previous authors [8-10] the maximum crack tip temperature increases were anticipated to be in the range of 100°C to 1000°C. At such temperatures the emitted infrared radiation is primarily in the 1 μ m to 10 μ m wavelength range. In addition, rise times for the temperatures were expected to be in the 1 μ s to 2 μ s range.

Our choice of indium antimonide (InSb), infrared (IR) detectors was dictated by the wavelength and rise time requirements described above. These detectors are sensitive to radiation with wavelengths from 1 μ m to 5.5 μ m. The particular system chosen consists of eight .16 mm \times .16 mm InSb detector elements in a linear array with a center to center spacing of 0.20 mm. The array is mounted in a liquid nitrogen (LN2) dewar to cool the detectors to 77K, thus minimizing their electrical noise and maximizing their sensitivity.

Each detector is connected to its own amplifier and the amplified signals are recorded on separate high speed, digital oscilloscopes at a 2 MHz sampling rate. The frequency response of the system of detectors and their amplifiers is from 4 Hz to 300 kHz. The 300 kHz upper limit corresponds to a minimum detectable rise time of .75 μ s, well below the anticipated 1 to 2 μ s rise times. Temperature increases as low as 20°C could be detected. However, when the temperature increase was less than approximately 50°C, the signal to noise ratio was on the order of one. Thus accurate data could not be obtained from surfaces at temperatures below 50°C above ambient.

Infrared optics

To measure the crack tip temperature distribution, the infrared radiation must be imaged from eight areas on the specimen surface onto the eight elements of the detector array. The optical system illustrated in Fig. 2 consists of two spherical mirrors and a plane mirror, and was used to focus radiation onto the detectors. The magnification of the optical system is one, and thus the size and spacing of the areas of measurement are equal to that of the detectors. As shown in Fig. 2, the size of the measurement areas is 160 \times 160 μ m, the center to center spacing is 200 μ m while the total length of the array from end to end is 1.56 mm.

Calibration

Careful calibration of the system is crucial for obtaining accurate results. The procedure was to heat a sample of the 4340 steel, with the same surface finish as the actual test specimens, in a furnace. The heated sample was removed from the furnace and was placed in front of the optical system such that the specimen was in focus. As the sample cooled the voltage output from the detectors and the sample temperature (measured with a thermocouple) were recorded simultaneously. It was found that the calibration data falls onto a straight line when plotted on a log-log scale.

Experimental results

Figure 3 shows the time record of the voltage output of each of the eight detector elements, as a crack approaches and passes through the detector points in specimen 9. Time $t = 0$ corresponds to triggering of the oscilloscopes. The material used in this experiment was an oil quenched 4340 steel whose material

characteristics appear in Table 1. For this experiment and for this particular measurement location, the crack tip speed was approximately constant and equal to 900 m/s. The maximum voltage increase was recorded by element 4 (Ch4.out in the figure). As is clear from the time record corresponding to this channel, the voltage increases rapidly from zero to a maximum value of 16.5 V in approximately 2 μ s. This rise time is substantially longer than the 0.75 μ s minimum rise time that can be accurately detected by our measuring system.

The voltage output time histories were converted into temperature histories by means of the calibration relation. Since this relation is non-linear, the resulting temperature records, shown here in Fig. 4, assume a slightly different shape than the voltage records. In this experiment the crack tip traversed the array of detection points slightly off center, but through the region focused on element 4. This element recorded a maximum temperature increase of 450°C while the rise time was 2 μ s. As may be expected by local symmetry, the elements to the left and right of element 4 (channels 3 and 5) recorded temperature increases which are very similar to each other. These elements also had markedly slower rise times than element 4.

Two alternative ways of viewing these results are shown in Figs. 5 and 6. These figures show contours of equal temperature increase near the propagating crack tip. The contours were obtained from the temperature increase versus time results of Fig. 4, by converting the time axis into distance parallel to the direction of crack growth, using the measured crack tip speed of 900 m/s and assuming that the crack propagates with a constant velocity.

In both Figs. 5 and 6 the estimated crack tip position is at $x_1 \approx -0.5$ mm, $x_2 \approx 0.6$ mm. The length scales of Fig. 5 show that this figure corresponds to a close-up view of temperature increase in the near vicinity of the propagating crack. The isotherms of Fig. 5 clearly show that intense heating (temperatures ranging from 450°C-150°C) extends in a narrow region approximately 1 mm ahead of the crack tip, while the half width of the resulting wake of temperatures is approximately 0.25 mm. It should be observed that the isotherms in the wake region behind the crack tip remain almost parallel to the crack line for at least 1.5 mm, suggesting that locally, the deformation remains essentially adiabatic (i.e., no substantial cooling by conduction is observed in this scale).

The scale of observation of Fig. 6 is very different. Here we present contours of temperature increase for distances behind the crack tip as large as 60 mm. Here the effect of cooling by conduction becomes apparent by the clear divergence of the isotherms behind the crack tip.

Note that the measurement are performed over small but nevertheless finite areas (see Fig. 2) and that the above contours have been obtained by assigning the recorded temperature history to the center of these areas. It should therefore be recognized that a certain amount of averaging is involved in this measurement. Nevertheless it is expected that the small size areas allow for the accurate investigation of the general structure of the temperature distribution near the tip. On the other hand, this unavoidable averaging makes the recorded maximum temperature increase a lower bound to the actual maximum temperature increase.

Discussion of results

Plastic work rate

The distribution of plastic work rate near the crack tip can be obtained by straightforward differentiation of the temperature field. Assuming that the heating due to plastic work is much higher than the thermoelastic effects and treating the crack growth as steady state, the heat conduction equation can be written as

$$k \nabla^2 T + f \sigma_{ij} \dot{\epsilon}_{ij}^P = - \rho c \dot{a} \frac{\partial T}{\partial x_1}, \quad (1)$$

with respect to a coordinate system translating with the crack tip where x_1 is the direction of crack growth, k is the thermal conductivity, T is the temperature, ρ is the density, c is the specific heat, \dot{a} is the crack speed, and f is the fraction of plastic work dissipated as heat.

Note that the conduction term, $k \nabla^2 T$, in equation (1) involves derivatives of T with respect to x_3 , while the measurement provides only $T(x_1, x_2, x_3 = h/2)$ where h is the specimen thickness. The very short time scales of this measurement (2-10 μ s) ensure that convection and radiation heat losses from the surface are very small. In addition conduction of heat from the interior of the specimen has no time to affect the surface temperatures near the crack tip. As a result, the variation of T with respect to x_3 is expected to be small, and little error is incurred by neglecting the $\frac{\partial^2 T}{\partial x_3^2}$ term in equation (1). Indeed when differentiating $T(x_1, x_2, x_3 = h/2)$ the entire conduction term was found to be small relative to the adiabatic heating term, $-\rho c \dot{a} \frac{\partial T}{\partial x_1}$.

The plastic work rate, $\dot{W}^P(x_1, x_2) = \sigma_{ij} \dot{\epsilon}_{ij}^P$, was obtained for all specimens assuming $f=0.9$ in equation (1). The results presented in Figs. 7 and 8 show the plastic work rate distribution for specimen 9. Figure 7 shows the contours of plastic work rate. The maximum value of the plastic work rate was $600 \times 10^{12} \text{J/m}^3 \text{s}$. It is seen that $\dot{W}^P > 0$, the region where the active plastic zone is just ahead of the crack tip and is elongated in the x_1 direction. The distribution is very symmetric for this particular experiment. The peak of \dot{W}^P which lies ahead of the crack tip is clearly seen in Fig. 8 where \dot{W}^P is plotted three dimensionally. This figure shows that except for a few oscillations, \dot{W}^P is essentially zero everywhere except in a region just ahead of the crack tip.

A rough estimate of the maximum plastic strain rate may be obtained from a simple calculation that assumes a von Mises yield criterion with a constant yield stress σ_0 equal to the static value given in Table 1 for the quenched and tempered steel. In this case

$$\| \dot{\epsilon}_{ij}^P(\underline{x}) \| = \frac{\dot{W}^P(\underline{x})}{\sigma_0} \sqrt{\frac{3}{2}}. \quad (2)$$

By using the maximum value of the plastic work rate, $\dot{W}_{\text{max}}^P = 600 \times 10^{12} \text{J/m}^3 \text{s}$, (2) gives maximum strain rate $\| \dot{\epsilon}_{ij}^P \|_{\text{max}} \approx 7 \times 10^5 \text{s}^{-1}$.

Role of Heating on Fracture Toughness

The experimental data show that temperature increases up to 465°C may occur at the tip of dynamically propagating cracks in 4340 steel. Such large temperature increases may have a significant effect on dynamic fracture propagation toughness. Three interacting phenomena that might be affected by local crack tip heating are outlined below. These are the decrease in yield stress at higher temperatures, the change in the fracture mechanism, and the change in the local structure of the near tip fields.

Data from the Metals Handbook on yield stress versus temperature for the tempered 4340 steel indicate that yield stress in tension at 450°C is 50% lower than that at room temperature. Dislocation motion is thermally activated and will be affected by local crack tip heating even on the time scales of this process, since there is no delay-time associated with the effect of temperature increase on mobility. As a result, near the crack tip there is a possibility of a substantial decrease of yield stress due to local heating.

A reduced yield stress near the crack tip will cause a change in the crack tip stress and strain fields compared to the fields that would exist under isothermal conditions. If crack growth is governed by a local criterion such as the attainment of a critical strain or opening angle at some distance near the crack tip, the effect of temperature on the local deformation fields will affect the fracture toughness of the material. It is not clear whether such effects will enhance or degrade fracture toughness. In general, lower yield stress values are associated with higher fracture toughness. Although this is true for a limited temperature range, this cannot be extrapolated indefinitely, since in limit as the temperature approaches the melting point, both σ_0 and the fracture toughness must eventually approach zero.

Data of Chi et al. [11] show that dynamic fracture initiation toughness for the tempered 4340 steel at 100°C is 50% higher than at room temperature. Care must be taken in extrapolating this data to apply to dynamic fracture propagation toughness since in performing high temperature fracture initiation tests the material will be held at a high temperature for some time, causing a certain amount of heat treating of the material to occur. However, in dynamic crack growth experiments there is insufficient time for any diffusion of atoms to take place and thus no heat treating can occur.

TABLE 2 - Summary of Experiments

Specimen	Heat Treatment	a, mm	\dot{a} , m/s	T_{max} , C	Notes
1	tempered	29	730	410	
5	tempered	29	980	415	
7	oil quenched	14	950	185	shear band temperature
9	oil quenched	14	900	450	
12	oil quenched	25	890	425	
13	oil quenched	8	1300	410	
17	oil quenched	8	1200	465	focused in circular groove
21	oil quenched	8	1100	≥ 420	Voltage went off scale
22	tempered	8	1900	290	focused in V-notch
24	tempered	8	1140	330	focused in circular groove

Conclusions

In an attempt to predict fracture toughness for dynamic crack growth, isothermal models use fracture criteria implemented very near the crack tip. It is exactly in such regions where the experimental data show temperature rises as high as 400°C. In light of the above observation, it appears that thermal effects should be incorporated in such models.

Acknowledgments

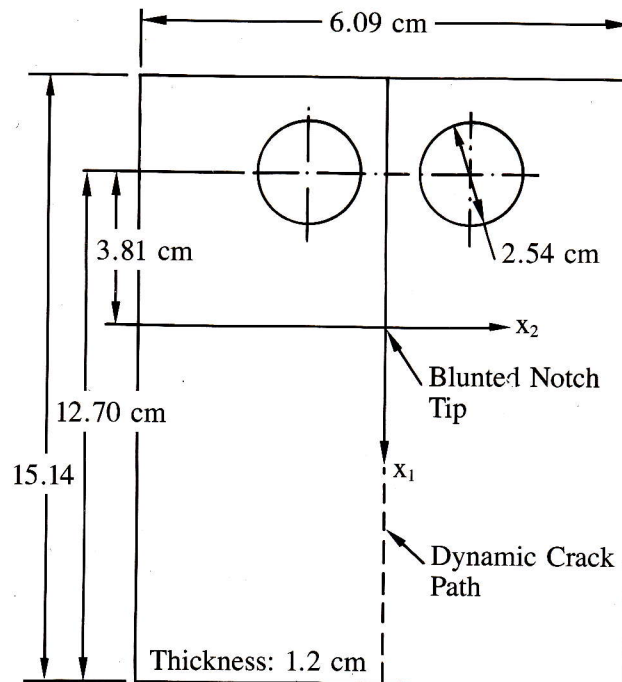
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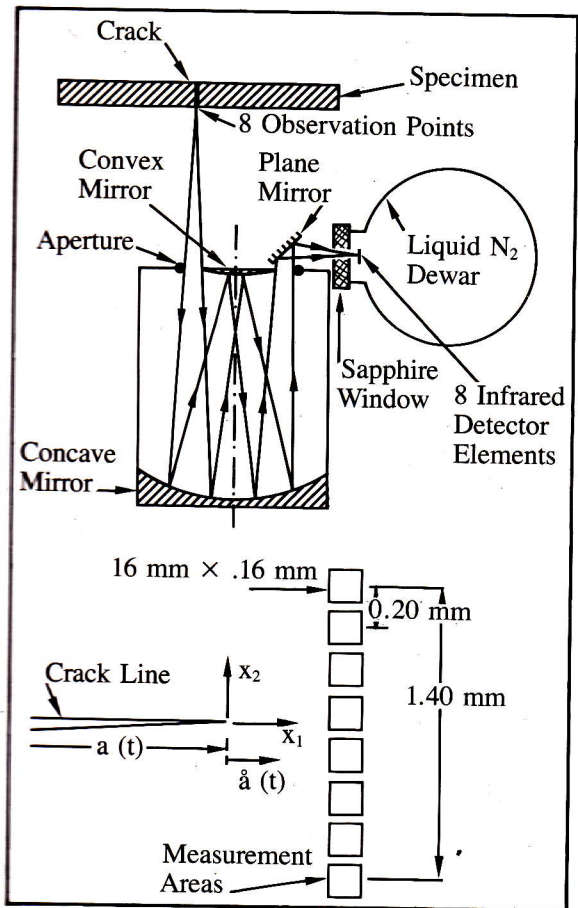


Fig. 2:

Schematic of focusing optics and of points on test specimen from which temperature is measured.

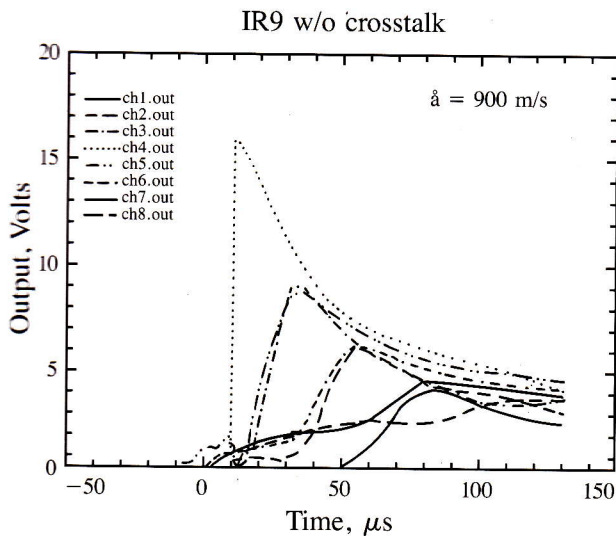


Fig. 3:

Specimen 9, oil quenched steel, IR detector output. Voltage versus time for each element.

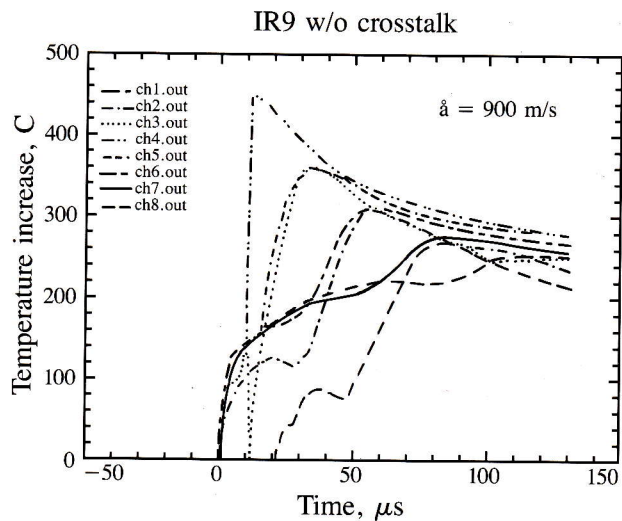


Fig. 4:

Specimen 9, temperature increase over ambient versus rise time for each element.

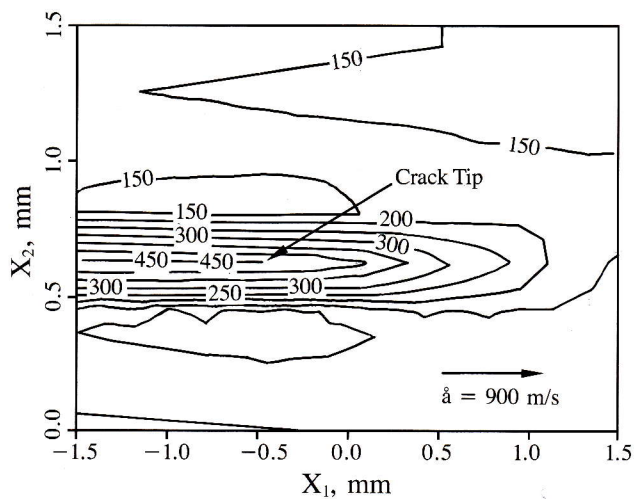


Fig. 5:

Specimen 9, contours of equal temperature rise very near the crack tip. Increment between contours is 25°C.

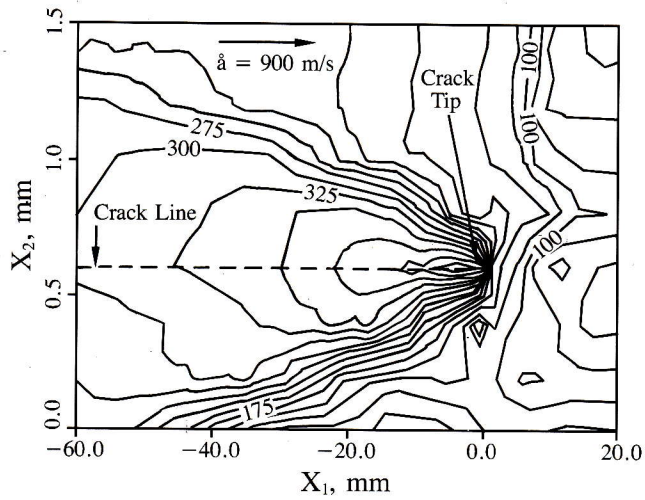


Fig. 6:

Specimen 9, contours of temperature increase over ambient in the region away from the crack tip.

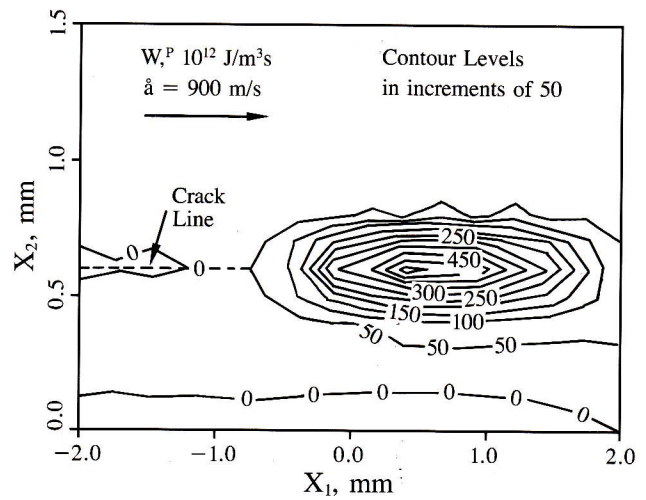


Fig. 7:

Specimen 9, contours of plastic work rate density near crack tip. Increment between contours is $50 \times 10^{12} \text{ J/m}^3\text{-s}$.

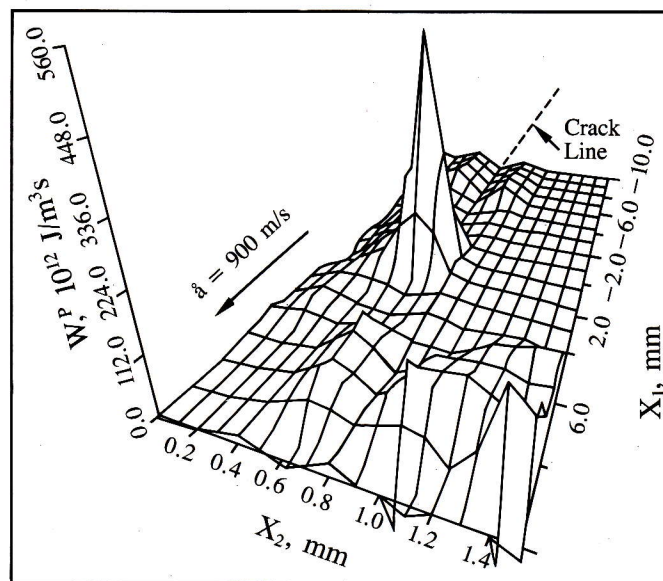


Fig. 8:

Specimen 9, surface plot of plastic work rate density.